# Synthesis and Characterization of Nanocrystalline Hydroxyapatite Gel and its Application as Scaffold Aggregation

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The sol-gel process is a technique used to synthesize materials from colloidal suspensions and, therefore, is suitable for preparing materials in the nanoscale. In this work hydroxyapatite was used due to its known properties in tissue engineering. Hydroxyapatite Ca<sub>10</sub>(PO<sub>4)6</sub>(OH), is a bioactive ceramic which is found in the mineral phase of bone tissue and is known for its great potential in tissue engineering applications. For this reason, this material can be applied as particle aggregates on ceramic slurry, coating or film on materials with a poorer biological response than hydroxyapatite. In this work, hydroxyapatite gel was obtained by the sol-gel process and applied as nanoparticle aggregation in the mixture of hydroxyapatite and tricalcium phosphate to form a ceramic slurry. This process is the polymer foam replication technique used to produce scaffolds, which are used in tissue engineering. For HA gel characterization it was used environmental scanning electron microscopy (ESEM), transmission electron microscopy (TEM), electron energy loss spectroscopy (EELS), scanning electron microscopy (SEM), X-ray diffraction (XRD) and X-ray fluorescence (XRF). The crystallite size was calculated from XRD data using the Scherrer equation. The nanoparticles size before firing was approximately 5nm. The crystallite size calculated after calcination was approximately 63 nm. The EELS results showed that calcium phosphate was obtained before firing. After HA gel calcination at 500 °C the XRD results showed hydroxyapatite with a small content of beta-TCP. The scaffolds obtained by polymer foam replication technique showed a morphology with adequate porosity for tissue engineering.

**Keywords:** nanoparticles, thin film, hydroxyapatite, scaffolds

#### 1. Introduction

Hydroxyapatite (HA) is a bioceramic material usually used in tissue engineering as bone substitute. For this application, HA is typically used in the form of powder or as scaffolds. In the latter case, HA should present a three-dimensional structure with interconnected pores, which guarantees desired similarities with bone tissue, besides being bioactive<sup>1</sup>.

Hydroxyapatite has been widely studied for its application as biomaterial. Moreover, recent efforts in the development of nanostructured hydroxyapatite materials have increased its applicability range. For example, calcium phosphate ceramic composites, such as hydroxyapatite/tricalcium phosphate (TCP), have been synthesized in such a way that the appropriate amounts of TCP and HA in the composite ensures the balance between the absorption and the formation processes of new tissue<sup>2</sup>.

In order to prepare nanostructured HA, different methods of synthesis can be used such as precipitation, sol-gel, microemulsions, sucrose-based route, among others. The sol-gel process can be defined as the conversion of a precursor solution (the sol, which is a colloidal suspension of solid particles dispersed in a liquid) into an inorganic solid via inorganic polymerization reactions induced by water. Usually, the precursor is a organic metallic compound such as alkoxides or an inorganic metal salt (for example: nitrate)<sup>3</sup>. This method can be performed by two different routes: Colloidal metal alkoxide<sup>4,5</sup> and polymeric<sup>6</sup>.

Colloidal metal alkoxide (or metal salt) route: The solvent can be alcohol or water and the resulting products are colloids in aqueous medium. Agglomerated powder may be formed after firing, which can be prevented by particle repulsion. This is the route adopted in this work.

**Polymeric route:** The solvent can be ethanol and the gel is formed by the condensation of small-sized particles. Amorphous structures can be formed, creating microporous systems.

In this work, the sol-gel process was used in order to produce homogeneous ceramic particles. This allowed the use of common reagents, besides being a relatively simple synthesis process.

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The sol-gel method<sup>7,8</sup> is used for the synthesis of powders, monoliths and thin films is based on the hydrolysis and polymerization (or condensation) of alkoxide molecular precursors. An advantage of the process is that they are synthetic and able to generate solutions of high purity.

Furthermore, the sol-gel process involves the use of liquid reagents with low viscosity and therefore good homogenization of the solution can be achieved in a short time.

Metal alkoxides are members of the family of metallorganic compounds, which have organic ligands that have bound to metal atoms. The ease of reaction between the metal alkoxides and water is the main property of these chemical compounds that are of direct interest in sol-gel process.

The production of the gel is divided into two stages: hydrolysis and condensation.

Hydrolysis is the main chemical reaction that leads to the transformation of precursor to final products (oxides). A variety of physical and chemical factors such as temperature, pH and concentration of chemical reagents, influence the polymerization process and, consequently, the properties of the oxides.

When it is required the synthesis of large quantities of material, water is preferred than ethanol, because required amount of the water and the cost are lower.

The nanostructures (scaffolds) are based on bioinspired design by manipulating and assembling of molecules and atoms forming the natural tissue. The scaffolds are formed by nanoparticles that are known for their high reactivity in the medium due to high surface area, consequently the nanoparticle assists in the formation of natural tissue. The large surface area of a nanoparticle often leads to distinct values of reactivity, mechanical, chemical and biological properties in vitro and in vivo when compared to micro and macroscopic particles<sup>9</sup>. Agglomeration of small particles is due to Van der Waals forces, this usually occurs in order to reduce the total surface energy. Van der Waals forces are weak, and it extends over a few nanometers. To reduce the Van der Waals interactions, repulsive forces must be present, which can be accomplished such as by electrostatic repulsion or steric hindrance.

When a hydroxyapatite gel layer is applied on the surface of a material (for example, a prosthesis), the dispersed nanoparticles adhere to the surface and this fact improves the contact of the prosthesis with the natural tissue, because the smaller size of the particles increases the surface area of the prosthesis. The larger region of interaction between materials improves the cell growth process.

The small size of nanoparticles facilitates the coating of complex surfaces when using, for example, the wet sponge technique.

The objective of this study was to characterize the hydroxyapatite gel that it is used for scaffolds coating.

## 2. Material and Methods

Hydroxyapatite gel was prepared using the water-based sol-gel method for synthesis. The reagents used was phosphoric acid 85% aqueous solution (LAFAN), calcium nitrate tetrahydrate (99% purity, Synth) and water (purified by reverse osmosis)<sup>10-12</sup>. The molar ratio Ca/P was 1.67.

In order to produce hydroxyapatite with Ca/P molar ratio = 1.67, for each 10 mol of calcium, 6 mol of phosphorous were used. The solvent was distilled water (H<sub>2</sub>O)<sup>13-15</sup>.

A commercial Hydroxyapatite - HA: Hidroxiapatita Fluidinova nanoXIM-HAp202 product 50120208 was used.

In order to produce tricalcium phosphate with Ca/P molar ratio = 1.5, for each 3 mol of calcium, 2 mol of phosphorous were used. The solvent was distilled water (H<sub>2</sub>O)<sup>16,17</sup>.

A commercial Tricalcium Phosphate - TCP: Fluidinova nanoXIM-TCP202 product 50220208 was also used.

The polymer foam replication technique was applied with ceramic slurry (2 g of HA, 2 g of TCP and 2 mL HA gel).

**First test** (commercial HA/TCP with HA gel): the sponge was submitted at a temperature of calcination of 725, with heating from 25 °C to 500 °C, and kept in 500 °C for 3 hours in order to eliminate the sponge. After this, the sponge is heated at 725 °C to finish the process.

**Second test** (synthesized HA/TCP with HA gel): the temperature of sinterization was 1330 °C with heating temperature from 25 °C to 500 °C, and kept in 500 °C for 3 hours in order to eliminate the sponge. After this, the next step was to increase the temperature up to 1330 to finish the process.

The sol-gel reaction is the transformation of the molecular precursor into a highly crosslinked solid and the process to obtain the gel is shown in Figure 1.

The process starts with (i) mixing two principal solution to form the sol; (ii) aging of the sol; (iii) drying the sol until the formation of a high viscosity gel, (iv) calcinating at 500 °C with heating rate of 3 °C per minute. Finally the powders were homogenized in agate mortar to eliminate possible agglomerates of nanoparticles (Figure 1).

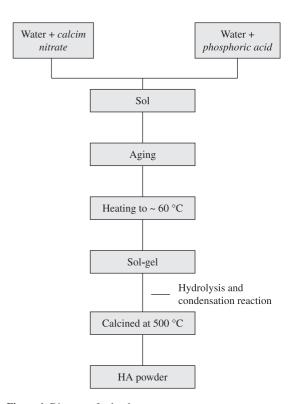


Figure 1. Diagram of sol-gel process.

In this work the X-ray diffraction data were analyzed using the *Scherrer* equation in order to determine the average crystallite size of the sample<sup>18,19</sup>. To describe the equation "t" <sup>20</sup>:

$$t = K \lambda \beta \cos \theta \tag{1}$$

Where  $\beta$  is the line width of the pure diffraction profile resulting from small crystallite size,  $\lambda$  is the X-ray wavelength, and  $\theta$  is the diffraction angle. K is the constant approximately equal to unity and related to crystallite shape. The Scherrer constant depends on how the width is determined, the shape of the crystal, and the size distribution. The most common values for K is 0.94 for FWHM (full width at half maximum) of spherical crystals with cubic symmetry. The parameter  $\beta$  was corrected by  $^{20}$ :

$$\beta = \beta_{\text{exp}}^2 - \beta_{\text{int}}^2 \tag{2}$$

Where  $\beta_{exp}$  is the diffraction line width and  $\beta_{inst}$  is the default width. The pattern is usually a single crystal with known crystallite size (Si or Al<sub>2</sub>O<sub>3</sub>). The parameters of single crystal used were  $\omega$ = 0.06784 and 20= 28.34989°. For this calculation it is considered that the X-ray diffraction peak can be described as a Gaussian distribution.

ESEM (environmental scanning electron microscopy) is used in the analysis of non-metallic surfaces because of accumulation of electric charges, and allows the evaluation of viscous liquids, gels, biological samples, among others<sup>21</sup>. This type of electron microscope neutralizes the negative charge on the sample surface when the beam interacts with the gas. The type of gas can be varied when needed. In this type of microscopy, the coating is unnecessary.

ESEM provides new tools to examine the dynamic behavior of various materials under different conditions<sup>22</sup>. The equipment used was FEI Quanta model 400 FEG ESEM with secondary electron detector to visualize the gel structure and morphology of nanoparticles.

The equipment used for nanoparticle analysis before calcination at 500 °C was the transmission electron microscope (Zeiss model CEM-902) with EELS analyzer, located at IQ-UNICAMP.

The samples were analyzed using a scanning electron microscope provided with a field emission gun, SEM-FEG, (JEOL JSM 6330F). In this case the samples were coated with carbon to visualize their morphology and particle size.

Samples' crystalline phases and crystallite size were determined by X-ray powder diffraction (XRD) using a Rigaku DMAX2200 diffractometer with Cu-K $\alpha$  radiation ( $\lambda$ = 1.5406Å) and Ni filter. The diffractograms were obtained with velocity of 0.01° s<sup>-1</sup> at a 2 $\theta$  range varying from 20° to 37°.

Impurity quantities were determined using a X-Ray Fluorescence (XRF) Rigaku RIX-3100 spectrometer. The samples were conformed into 25 mm disks and analyzed by a semi-quantitative routine.

The Nanoparticle Tracking Analysis (NTA) instrument (NanoSight NTA 2.0 model LM20) comprise of a scientific camera, microscope and a sample-viewing unit and are used in conjunction with a computer control unit, which runs the NTA software. The equipment uses a 638 nm red laser diode to illuminate particles in liquid suspension, which are held within the unit. NTA is used to analyze videos captured by the instrument, giving a particle size distribution and particle count.

The software identifies and tracks individual nanoparticles which are moving with Brownian motion and relates the movement to a particle size. This technique analyzes particles in liquids from 10 to 1000 nm (depending on material).

#### 3. Results and Discussion

## 3.1. Nanoparticle analysis

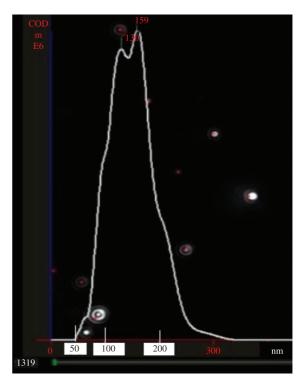
Because of the gel struture, there were no particles moving under Brownian motion. After dilution (about 500 times) in pure water (18 M Ohm) the results obtained are shown in Figure 2. The system was polydispersed and all particles were drifting (might be due to the destruction of the gel structure).

Figure 2 shows particle agglomerates of about 150 nm. These particle agglomerations also are shown in Figure 3 obtained by ESEM.

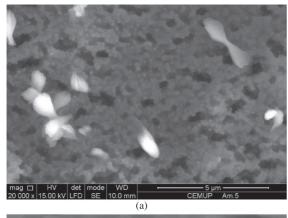
## 3.2. Environmental scanning electron microscopy and scanning electron microscopy

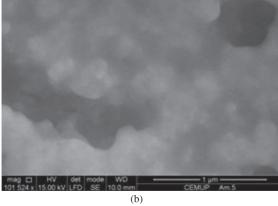
In Figure 3a it is possible to verify the gel morphology with porous structure, obtained by ESEM. In this sample, coating was not made in order to avoid hiding the real size of particles.

In Figure 3b it is possible to verify the gel and nanoparticles. The gel isolates the nanoparticles, and according with Figure 3a, b, it can also be observed that the nanoparticles are spherical. Due to the structure of the gel, the image resolution was not perfect, since the sample was coated by plasma sputter coater in order to improve the conduction of electrons in the sample. By comparison with the scale it can be observed particles with size around 100 nm.



**Figure 2.** Nanoparticle analysis result (Nanosight). The particle size agglomerates is about 150 nm.





**Figure 3.** (FEG ESEM) Hydroxyapatite gel before calcination. (a) Overall dispersion of particles in the gel and some pores. (b) Particles dispersed in detail.

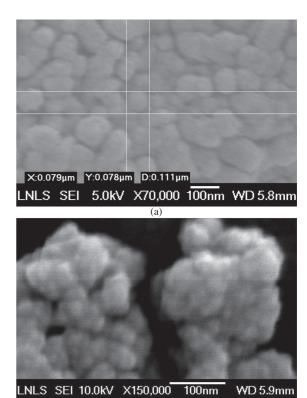
The hydrolysis during the heating process leads a dispersion of colloidal particles in a condensed gel with an interconnected and porous network. This step of the process is called the sol-gel transition and is shown in Figure 3b. The results show strong similarities with the theory of the structural glass transition<sup>23</sup>. The gelling solution at the sol-gel transition is a unique state of matter that is neither liquid nor solid, but rather is in transition between these states<sup>24</sup>.

The resulting structure after the gel formation and shown in the ESEM image (Figure 3) is similar with the theoretical images obtained by colloidal route<sup>25,26</sup>.

When the gel is dried under ambient condition a solid material called xerogel is formed. This type of structure normally retains high porosity and large surface area and their pores are small, usually in the nanoscale<sup>27</sup>.

In Figure 4a, it is possible visualize the morphology of agglomerates and the material structure. In the detail it can be seen a nanoparticle aggregate with approximately 111 nm.

Figure 4b shows images of nanoparticles with size less than 50 nm. It was thought that the particles that is observed in Figure 4b was the gold coating, but the gold particles have a size of about 5 nm, which is 10 times smaller than the particles shown in Figure 4b<sup>28</sup>.



**Figure 4.** (a) Morphology of the particle aggregates and the cluster size of particles (500 °C). (b) Particles agglomerates (SEM-FEG).

#### 3.3. TEM and EELS

Figure 5 shows dispersed nanoparticles with sizes smaller than 5 nm, but it can also be seen nanocrystals with approximately 20 nm in diameter.

We believed that when the solution becomes more concentrated and viscous, the nanoparticles of 5 nm are aggregated, forming larger crystals as shown in some crystals in this image. After calcination, the size of nanoparticles increased as shown in Figure 4, with nanoparticles between 80 nm to 100 nm.

Excitation counts of Ca  $L_{3,2}$ -edge (Figure 5) with variation of energy loss was found to be between of 340-360 eV<sup>29</sup>. According to the EELS atlas, calcium (Ca) has the  $L_{3,2}$ -edges and have a loss of energy 346.4 eV.

## 3.4. XRD - X-ray diffraction

Figure 6a shows the XRD of hydroxyapatite gel calcined at 500 °C for 4 hours. This sample shows the major peak of hydroxyapatite in 31.6° that corresponds to hydroxyapatite (211) plane<sup>30</sup>, also between 30° and 31° there is a peak with low intensity of beta-tricalcium phosphate phase.

The parameters used in Scherrer equation were obtained and used to calculate a crystallite size of 63 nm. This value analyzing was obtained by the main diffraction peak (Figure 6b) ,which was in a diffraction angle  $2\theta$  of  $31.56621^{\circ}$  (xc l). The Gaussian parameter was  $\omega$ = 0.23275 (w1).

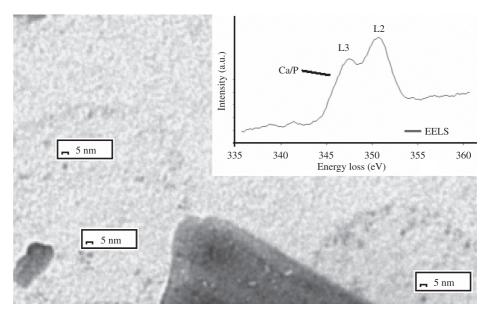
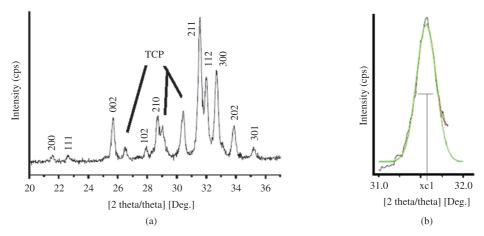


Figure 5. Results of HA gel before calcination obtained by transmission electron microscope. Detail: nanoparticles with approximately 5 nm.



**Figure 6.** (a) XRD of hydroxyapatite gel calcined at 500 °C showing the HA crystal plane and some peaks of tricalcium phosphate - TCP. (b) Gaussian curve of the highest intensity peak of hydroxyapatite.

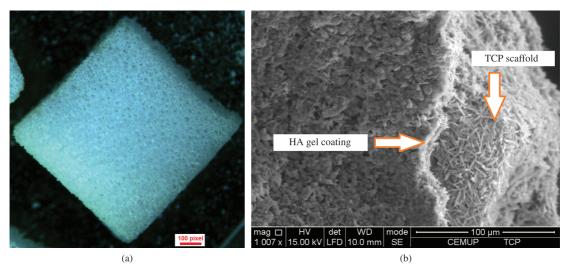


Figure 7. (a) Commercial HA/TCP ceramic sponge with HA gel coating calcined at 725 °C. (b) Detailed picture of the fracture where it is possible to verify the presence of HA gel coating.

Table 1. Results of XRF semi-quantitative analysis (wt. (%)).

	Sr	Na	S	Fe	Mg	Si	Ni
500 °C	0.028	0.013	0.004	0.004	0.004	0.003	0.002

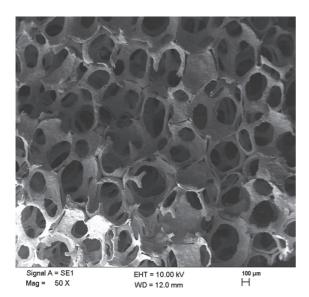


Figure 8. Synthesized HA/TCP ceramic sponge obtained by slurry ceramic, sintered at 1330 °C. Pores of scaffold.

## 3.5. XRF - X- ray fluorescence

The XRF that is a semi-quantitative chemical analysis and showed that HA has low level of impurities and do not have heavy metal contamination (Table 1). This is very important because biomaterials can not have this kind of chemical elements in their structure.

#### 3.6. Polymer foam replication technique

It is known that HA is a bioactive bioceramic and from the results of this work it can be seen good perspectives on the development of new orthopedic devices and tissue engineering scaffolds.

The scaffold (commercial HA/TCP) morphology calcined at 725 °C is shown in Figure 7a, Figure 7b shows the detail of ceramic scaffold fracture, where it is possible to see the HA coating and the scaffold nanostructure.

Figure 8 shows the scaffold prepared with synthesized HA/TCP and sintered at 1330 °C from slurry ceramic. The pores of the scaffold were found to be adequate to use in tissue engineering applications.

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#### 4. Conclusions

The gel was efficient to aggregate the particles by creating a resistant structure due to the fact that the particles have a stronger bond, because of the gel structure. This scaffold is very porous, so any increase on mechanical resistance is important.

The scaffolds obtained with commercial materials calcined at 725 °C presented nanostructure and it was possible to see the HA layer in the fracture of the ceramic sponge. Also, the sponge that was prepared using synthesized HA/TCP and sintered at 1330 °C presented interconnected pores and pores size adequate to tissue engineering applications.

Nanoparticles were already formed in the mixtures of the two solutions, but with particles size around 5 nm as shown in the TEM pictures. With stirring and heating, the gel became more viscous (hydrolysis and condensation), the particles tended to increase their size by agglomerating in some places, forming clusters around 150 nm. This was observed from the results of ESEM and NTA.

With the EELS it was proved the presence of calcium indicating the formation calcium phosphate (hydroxyapatite) before calcination. With XRD it was proved the presence of hydroxyapatite crystalline phase in the diffractogram after calcination.

XRD data was used based on the main peak to perform the calculation of crystallite size, where the average value obtained was 63 nm.

With SEM-FEG was proved that after calcination at  $500\,^{\circ}$ C the sample had it particle size agglomerates of about  $100\,\text{nm}$ .

The XRF showed that HA presented low level of impurities and did not present heavy metal contamination.

Due to the material characteristics obtained with XRD, SEM and XRF it was suggested that the scaffolds obtained in this work might be adequate for tissue engineering applications.

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